

Структура полимеров и композитов

Андрей Петухов

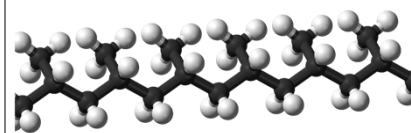
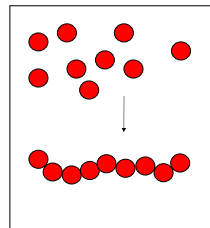
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What is a polymer?

- A long molecule made up from lots of small molecules called **monomers**.

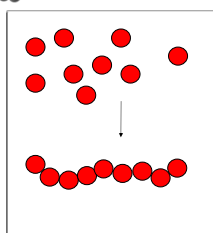
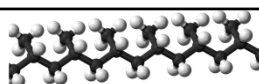


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What is a polymer?

typical energy scales:

- C-C bond 350 kJ/mol (≈ 3.5 eV)¹
 - At room temperature $k_B T = 2.5$ kJ/mol (≈ 25 meV)
- intermolecular interactions
 - Hydrogen bond
 - dipole-dipole
 - van der Waals
- 10-100 times weaker than covalent bonding



¹J. E. Huheey, E. A. Keiter, and R. L. Keiter, Inorganic Chemistry, 4th ed. (1993)

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Examples of Natural and Synthetic Polymers

Natural polymers are made by living organisms.

Synthetic polymers are made by chemical reactions in a lab.



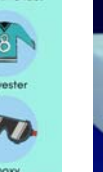
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Can you think of an application where polymers are not used?
I failed to find one...

Examples of Natural and Synthetic Polymers

Natural polymers are made by living organisms.

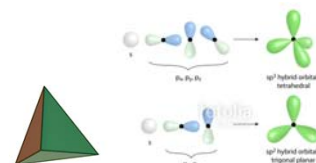
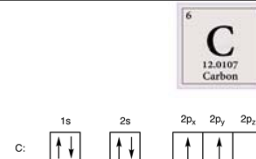
Synthetic polymers are made by chemical reactions in a lab.



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Carbon

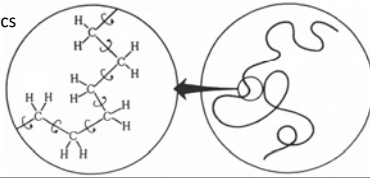
- Element #6
 - 6 electrons
- 1s² 2s² 2p²
- sp² hybridization: trigonal planar
 - example: graphene
 - bond angle 120°
- sp³ hybridization: tetrahedral 3D
 - examples: diamond, methane CH₄
 - bond angle 109.5°



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Example: Polyethene (=Polyethylene)

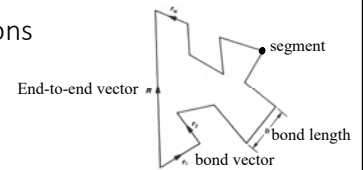
- in an ideal solvent
- undergoing random conformations
- bond angle 109.5°
- described by statistical mechanics
- entropy is very important



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Polymer conformations

$$\text{End-to-end vector } \vec{R} = \sum_{i=1}^N \vec{r}_i$$



$$\text{Average } \langle \vec{R} \rangle = 0$$

$$\langle \vec{R}^2 \rangle = \langle \vec{R} \cdot \vec{R} \rangle = \left\langle \sum_{i=1}^N \sum_{j=1}^N \vec{r}_i \cdot \vec{r}_j \right\rangle = \sum_{i=1}^N \langle \vec{r}_i^2 \rangle + \left\langle \sum_{i=1}^N \sum_{j \neq i}^N \vec{r}_i \cdot \vec{r}_j \right\rangle \rightarrow 0 \text{ for large } |j-i|$$

$$\langle \vec{R}^2 \rangle \propto N \Rightarrow \sqrt{\langle \vec{R}^2 \rangle} \propto N^{1/2} = Nb^2 \propto N$$

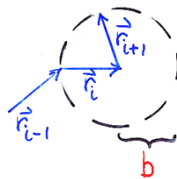
8

Chain models: freely jointed chain

$$\langle \vec{R}^2 \rangle = \sum_{i=1}^N \langle \vec{r}_i^2 \rangle + \left\langle \sum_{i=1}^N \sum_{j \neq i}^N \vec{r}_i \cdot \vec{r}_j \right\rangle$$

$$\langle \vec{r}_i \cdot \vec{r}_j \rangle = \begin{cases} b^2, & i = j \\ 0, & i \neq j \end{cases}$$

$$\langle \vec{R}^2 \rangle = \sum_{i=1}^N \langle \vec{r}_i^2 \rangle + \left\langle \sum_{i=1}^N \sum_{j \neq i}^N \vec{r}_i \cdot \vec{r}_j \right\rangle = Nb^2$$



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Chain models: fixed bond angle

$$\langle \vec{r}_i \cdot \vec{r}_i \rangle = b^2$$

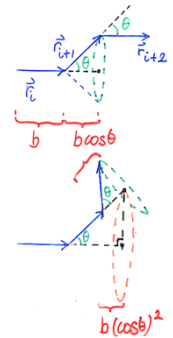
$$\langle \vec{r}_i \cdot \vec{r}_{i+1} \rangle = b^2 (\cos \theta)^1$$

$$\langle \vec{r}_i \cdot \vec{r}_{i+2} \rangle = b^2 (\cos \theta)^2$$

...

$$\langle \vec{r}_i \cdot \vec{r}_j \rangle = b^2 (\cos \theta)^{|j-i|}$$

For sp^3 hybridization: $\cos(180^\circ - 109.5^\circ) = 1/3$



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Chain models: fixed bond angle

$$\langle \vec{R}^2 \rangle = \sum_{i=1}^N \langle \vec{r}_i^2 \rangle + \left\langle \sum_{i=1}^N \sum_{j \neq i}^N \vec{r}_i \cdot \vec{r}_j \right\rangle \quad \langle \vec{r}_i \cdot \vec{r}_j \rangle = b^2 (\cos \theta)^{|j-i|}$$

$$\langle \vec{R}^2 \rangle = \sum_{i=1}^N b^2 [1 + 2(\cos \theta)^1 + 2(\cos \theta)^2 + 2(\cos \theta)^3 + \dots] +$$

$$\langle \vec{R}^2 \rangle = Nb^2 + 2Nb^2[(\cos \theta)^1 + (\cos \theta)^2 + (\cos \theta)^3 + \dots]$$

$$\langle \vec{R}^2 \rangle = Nb^2 \left[1 + 2 \frac{\cos \theta}{1 - \cos \theta} \right] = Nb^2 \frac{1 + \cos \theta}{1 - \cos \theta}$$

$$\langle \vec{R}^2 \rangle = Nb_{\text{eff}}^2 \text{ with } b_{\text{eff}} = b \sqrt{\frac{1 + \cos \theta}{1 - \cos \theta}} \quad \text{For } sp^3 \text{ hybridization: } b_{\text{eff}} = b\sqrt{2}$$

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Diffusion analogy

Cf. one diffusing colloidal particle or molecule (Einstein):

$$\langle \vec{R}^2 \rangle = 6Dt$$

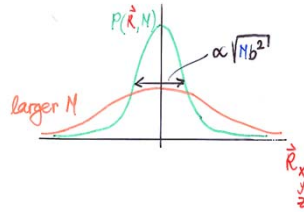


An (ideal) polymer is like the **trajectory** of a diffusing particle!

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End-to-end distance distribution

$$P(\vec{R}, N) = \left(\frac{3}{2\pi Nb^2} \right)^{3/2} \exp\left(-\frac{3\vec{R}^2}{2Nb^2} \right)$$



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Central Limit Theorem

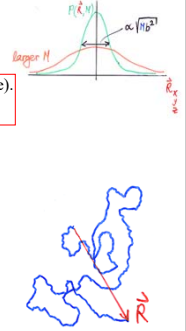
Consider the sum of N independent, stochastic variables (N is large).
This sum has a normal (= Gaussian) distribution with $\sigma^2 \propto N$.

$$\vec{R} = \sum_{i=1}^N \vec{r}_i \Rightarrow \sigma_{\vec{R}}^2 = \left\langle (\vec{R} - 0)^2 \right\rangle = \left\langle \vec{R}^2 \right\rangle \propto N \left(\times b_{eff}^2 \right)$$

$$\text{Variation in } \vec{R}^2 : \sigma_{\vec{R}^2} = \sqrt{\left\langle \left(\vec{R}^2 - \left\langle \vec{R}^2 \right\rangle \right)^2 \right\rangle} = \sqrt{\frac{2}{3} Nb^2}$$

A Gaussian coil is a strongly fluctuating object!

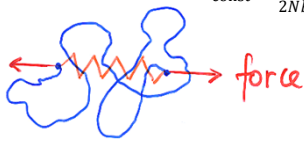
Conclusion: many (global) properties of polymers do not depend on the (local) details of the model.



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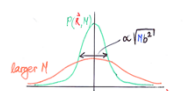
Entropy of a Gaussian coil

$$\begin{aligned} S(\vec{R}) &= k_B \ln W \\ &= \text{const} + k_B \ln P(\vec{R}) \\ &= \text{const} - \frac{3k_B}{2Nb^2} \vec{R}^2 \end{aligned}$$



$$P(\vec{R}, N) = \left(\frac{3}{2\pi Nb^2} \right)^{3/2} \exp\left(-\frac{3\vec{R}^2}{2Nb^2} \right)$$

$$A(\vec{R}) (= -TS) = \text{const} + \frac{3k_B T}{2Nb^2} \vec{R}^2 \quad \text{ENTROPIC SPRING}$$



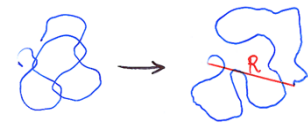
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Non-ideal polymer chains

$$U[c(\vec{R})] = Nk_B T B c(\vec{R})$$

B is the **second virial coefficient** ($B > 0$ repulsion)

$$\text{Edwards (1965): } \left\langle \vec{R}^2 \right\rangle \propto N^{6/5}$$



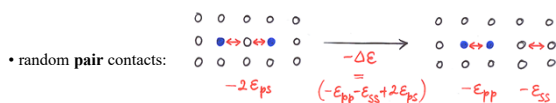
SHORT-RANGE REPULSION: swelling

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Repulsion combined with attraction

Assumptions:

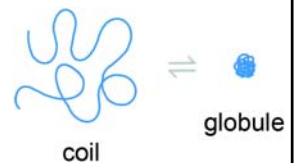
- only attraction between nearest neighbours
- coordination number: z
 - solvent-solvent $\circ \circ$: $-\epsilon_{ss}$
 - solvent-polymer $\circ \bullet$: $-\epsilon_{ps}$
 - polymer-polymer $\bullet \bullet$: $-\epsilon_{pp}$



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introduction to Flory-Huggins theory

Structure is controlled by the $\frac{z\Delta\epsilon}{k_B T} \equiv \chi$ (chi-parameter)



- at $\chi = 0$ swollen chain $R^* \propto N^{3/5}$
- at $\chi = 1/2$ (θ - temperature): ideal chain $R^* = R_0^* \propto N^{1/2}$
- at $\chi > 1/2$: collapsed globule, $R^* = R_0^* \propto N^{1/3}$
the transition is rather sharp for large N

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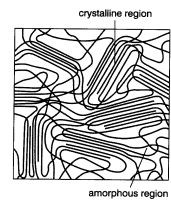
Going to concentrated solutions and polymeric solids

- there are multiple models
- too short time to discuss them here
- only a few concepts are discussed below

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Semi-crystalline polymers

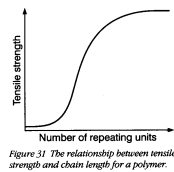
- Areas in polymer where chains packed in regular way.
- Both amorphous and crystalline areas in same polymer.
- Crystalline - regular chain structure - no bulky side groups.
- More crystalline polymer - stronger and less flexible.



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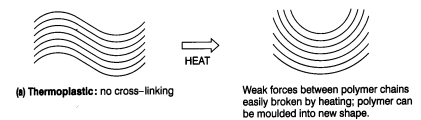
Chain length effect

- Critical length needed before strength increases.
- Hydrocarbon polymers average of 100 repeating units necessary but only 40 for nylons.
- Tensile strength measures the forces needed to snap a polymer.
- More tangles + more touching!!!



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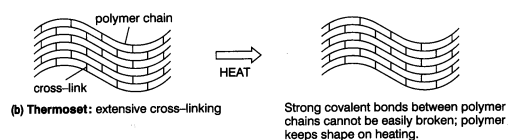
Thermoplastics (80%)



- No cross links between chains.
- Weak attractive forces between chains broken by warming.
- Change shape - can be remoulded.
- Weak forces reform in new shape when cold.

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Thermosets



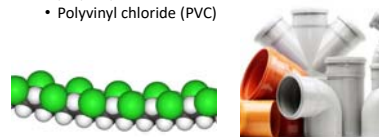
- Extensive cross-linking formed by covalent bonds.
- Bonds prevent chains moving relative to each other.

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Homopolymers

- All monomers are of the same type
 $A + A + A + A + A + \dots \rightarrow -A-A-A-A-$

- Examples:
 - polyethylene
 - Polyvinyl chloride (PVC)



Monomer type affects the intermolecular interactions but also...

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Polymer properties affected by

- chain length,
- presence and type of side groups,
- chain branching,
- stereoregularity,
- chain flexibility,
- crystallinity,
- cross linking...

• and this is not yet all!

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Homo- and hetero-polymers

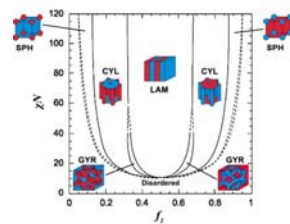
1. homopolymer
2. alternating copolymer
3. random copolymer
4. block copolymer
5. grafted copolymer



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Self-assembled structures of block co-polymers

- Appear for sufficiently high asymmetry parameter.
- Tunable structure.



A.H. Hofman, PhD thesis, RUG, 2016

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Composite materials

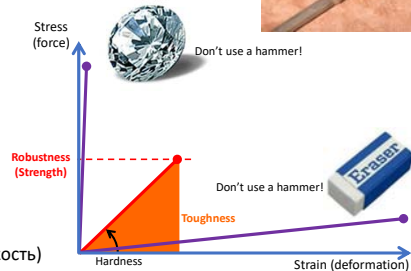
- consist of two or more constituent materials
 - e.g., reinforced concrete
 - plywood
 - fiber-reinforced plastics
 - ...



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Hardness, robustness & toughness

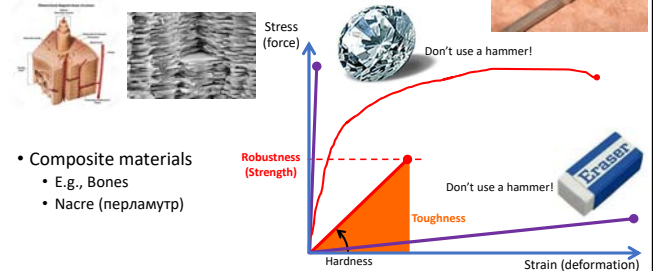
- Hardness (Твердость)
- Robustness (Прочность)
- Toughness (Ударная вязкость)



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Hardness, robustness & toughness

- Composite materials
 - E.g., Bones
 - Nacre (перламутр)



- Soft matter is usually not hard but composites can be strong and tough!

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Composite materials

- E.g., carbon nanofibers
 - sp^2 chemical bonds in graphite are stronger than in diamond
 - however, weak interlayer bonding
 - use nanofibers to reinforce polymers!



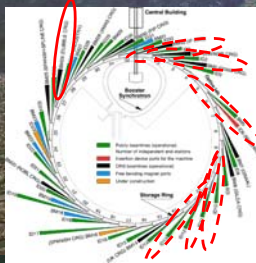
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Structural studies with x-rays



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ESRF, Grenoble (France)



Synchrotron operates at $E_e = 6 \text{ GeV}$

$$E_e = \gamma mc^2$$

$$\gamma = \frac{1}{\sqrt{1-v^2/c^2}}$$

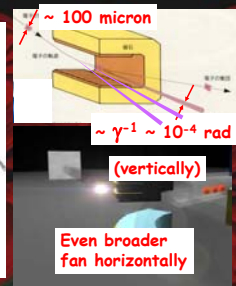
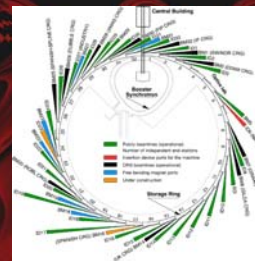
$$mc^2 = 0.5 \text{ MeV}$$

$$\gamma > 10^4$$

$$v = 0.999999996 \text{ c}$$

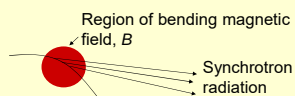
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Source at BM-26 'DUBBLE': bending magnet



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Synchrotron radiation: undulators



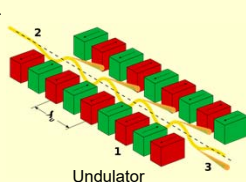
Electron beam of energy E

Characteristic energy, E_c

$$E_c = 0.665 B E^2 \text{ keV}$$

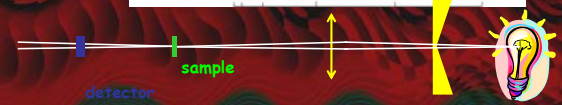
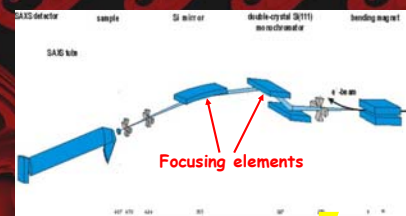
E is in GeV, B is in T

$$\lambda_c = 18.6 / B E^2 \text{ Angstrom}$$



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Optical scheme of DUBBLE beamline



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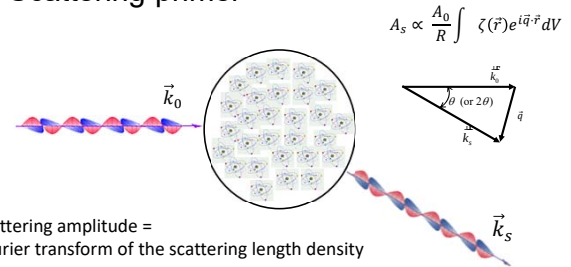
SynchrotronLIKE in Kaliningrad

- Small angular source size seen from the sample position
- Similar to that at 'large' synchrotron but lower brightness
- Perfect if no time resolution is needed



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Scattering primer

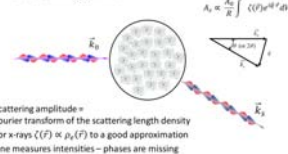


- Scattering amplitude = Fourier transform of the scattering length density
- For x-rays $\zeta(\vec{r}) \propto \rho_e(\vec{r})$ to a good approximation
- One measures intensities – phases are missing

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What do we measure?

Scattering primer



- Scattering amplitude = Fourier transform of the scattering length density
- For x-rays $\zeta(\vec{r}) \propto \rho_e(\vec{r})$ to a good approximation
- One measures intensities – phases are missing

$$I \propto |A_s|^2 = A_s \times A_s^* = \int \rho_e(\vec{r}) e^{i\vec{q} \cdot \vec{r}} d^3\vec{r} \int \rho_e(\vec{r}') e^{-i\vec{q} \cdot \vec{r}'} d^3\vec{r}' \quad \text{replace } \vec{r}' = \vec{r} + \delta\vec{r}$$

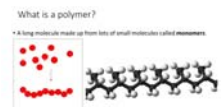
$$= \iint \underbrace{\rho_e(\vec{r}) \rho_e(\vec{r} + \delta\vec{r})}_{\text{Density correlation function}} d^3\vec{r} d^3\delta\vec{r}$$

→ Scattering measures density correlations

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Small-angle approximation

- Wavelength ~ 1 Angstrom = 0.1 nm
- Scale of interest 1 nm – 1 μm
- → (ultra) small angles



Ewald sphere

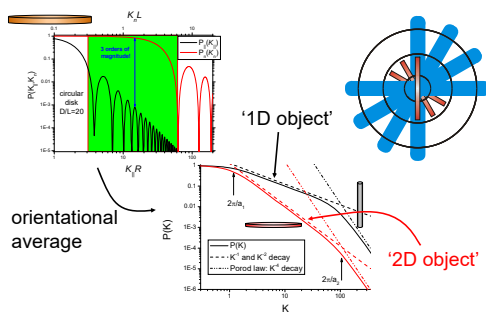
Ewald sphere is (almost) a plane

Scattering amplitudes are determined by a 2D cross section through the 3D reciprocal space

If structure is isotropic *in average*, scattering does not depend on the direction

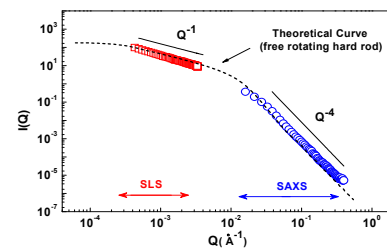
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Low-dimensional objects: disks & rods



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Sepiolite rods

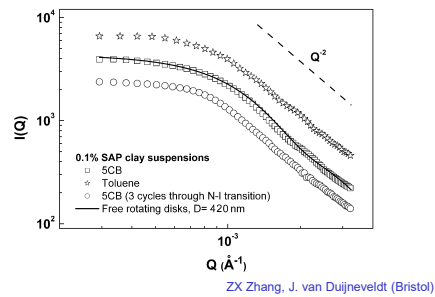


Theory: L = 900 nm (40% polydisperse) and D = 30 nm

ZX Zhang, J. van Duijneveldt (Bristol)

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Montmorillonite clay platelets – light scattering



Random polymer coils

Polymer in ideal solvent

- mass $M \propto$ contour length L
- $R_g \propto \sqrt{L} \propto \sqrt{M}$, where $R_g = \sqrt{6} R_{\text{end-to-end}}$
- $M \propto R_g^2$
- An ideal random coil has the mass dimension of 2
 - $\gg$ chain diameter
 - $\ll$ coil size
 - yields $I \propto q^{-2}$ decay



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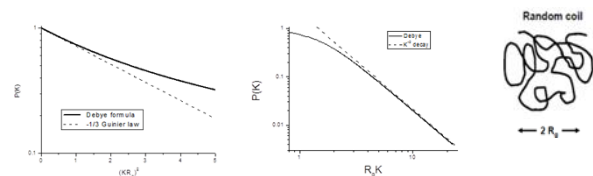
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Random polymer coils: Guinier limit

- Guinier limit:
 - small K
 - scales $\gg$ coil size
- Guinier law
 - $I(K) \propto \exp[-(R_g K)^2/3]$
 - $R_g = \sqrt{6} R_{\text{end-to-end}}$



Random polymer coils



$$P(K) = \frac{2}{(R_g K)^4} \left[(R_g K)^2 - 1 + \exp(-(R_g K)^2) \right]$$

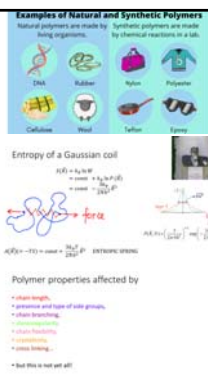
$$\exp(-(R_g K)^2) \approx 1 - (R_g K)^2 + \frac{1}{2}(R_g K)^4 - \frac{1}{6}(R_g K)^6 + \dots$$

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Conclusions

- Polymers are indispensable in modern life
- Strongly influenced by entropy
- Broad opportunities for materials engineering
- More dimensions with composites
- Scattering enlightens the nanostructure



Acknowledgement

- Thanks to
 - Gert Jan Vroege
 - www.worldofteaching.com
- Thank you for your attention!

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